镁掺杂氧化铈整体式催化剂催化 CO₂和 CH₃OH 直接合成碳酸二甲酯

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摘要:采用共沉淀法成功地合成了不同 Mg掺杂量的 Ce_{1-x}Mg_xO₂(x=0.05、0.10、0.15、0.20)固溶体催化材料,并运用透射电子显微 镜(TEM)、X 射线衍射(XRD)、氮气吸附-脱附测试、拉曼光谱、X 射线光电子能谱(XPS)、CO₂程序升温脱附(CO₂-TPD)等技术对这 些材料进行了表征。结果发现,通过调控 CeO₂晶格中 Mg 的含量,可以调控所制备的 Ce_{1-x}Mg_xO₂催化材料的粒径、比表面积、表 面缺陷等。其中 Ce_{0.90}Mg_{0.10}O₂展现了最佳的表面性质,具有最小的平均粒径(约 5.8 nm),最大的比表面积(约 136 m²·g⁻¹)以及最 高的表面氧含量(31.98%)。将 Ce_{1-x}Mg_xO₂催化材料涂覆在堇青石蜂窝陶瓷上制成整体催化剂,考察其对 CO₂和 CH₃OH 直接合 成碳酸二甲酯的催化性能。在 140 ℃、2.4 MPa、反应 2 h 的条件下, Ce_{0.90}Mg_{0.10}O₂整体催化剂上碳酸二甲酯的收率高达 20.21%, 催化效果明显优于 CeO₂和其余的 Ce_{1-x}Mg_xO₂(x=0.05、0.15、0.20)催化材料。

关键词: CO₂转化;碳酸二甲酯;氧空位;整体式催化剂;铈镁复合氧化物
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Direct Synthesis of Dimethyl Carbonate from CO₂ and Methanol by Mg-Doped Ceria Monolithic Catalyst

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Abstract: In this paper, $Ce_{1-x}Mg_xO_2$ (x=0.05, 0.10, 0.15, 0.20) solid solution catalytic materials with different molar ratios were successfully synthesized by co-precipitation method. These materials were characterized by transmission electron microscope (TEM), X-ray diffraction (XRD), nitrogen adsorption-desorption test, Raman spectroscopy, X-ray photoelectron spectroscopy (XPS), CO_2 temperature-programmed desorption (CO_2 -TPD) and other techniques. It was found that the particle size, specific surface area, surface defects, *etc.* of the prepared $Ce_{1-x}Mg_xO_2$ catalytic materials can be tuned by regulating the content of Mg in the CeO_2 lattice. Among them, $Ce_{0.90}Mg_{0.10}O_2$ exhibited the best surface properties, with the smallest average particle size of about 5.8 nm, the largest specific surface area of about 136 m² · g⁻¹, and the highest surface oxygen content (31.98%). $Ce_{1-x}Mg_xO_2$ catalytic material was coated on the cordierite honeycomb ceramic to make a monolithic catalyst, and its catalytic performance for the direct synthesis of dimethyl carbonate from CO_2 and CH_3OH was investigated. Under the conditions of 140 °C, 2.4 MPa, and 2 h reaction, the yield of dimethyl carbonate on $Ce_{0.90}Mg_{0.10}O_2$ monolith catalyst was as high as 20.21%, and the catalytic

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activity was significantly higher than that of CeO_2 and other $Ce_{1-x}Mg_xO_2$ (x=0.05, 0.15, 0.20) catalytic materials.

Keywords: CO2 conversion; dimethyl carbonate; oxygen vacancies; monolithic catalyst; magnesium-cerium oxides

0 Introduction

Dimethyl carbonate (DMC) has been widely applied as a fuel additive, in electrochemistry and organic synthesis due to its environmental-friendly properties^[1-3]. Although many methods have been applied for DMC synthesis, such as phosgene method, transesterification method, urea alcoholysis method, epoxy alkane method, methanol, and CO₂ direct synthesis method, etc.^[4-6]. Direct synthesis of DMC from CO₂ and methanol has attracted great attention (Scheme 1)^[7]. The utilization of CO₂ as the carbon source instead of fossil feedstock may promote the sustainability of the chemical industry and terminate the greenhouse effect caused by excessive CO₂ emission. However, there are still some vital challenges such as low yield, deactivation of the catalyst, and thermodynamic limitations for this route^[6,8]. Thus, designing novel catalysts and developing efficient water removing methods from the reaction mixture are crucial to overcoming the thermodynamic equilibrium of the reaction.

$$2CH_{3}OH + CO_{2} \xleftarrow{Catalyst} H_{3}CO OCH_{3} + H_{2}O$$

Scheme 1 Direct synthesis of DMC from CO₂ and methanol

Ceria-based nanomaterials have been widely studied in the direct synthesis of DMC from CO_2 and methanol. This is mainly due to its fascinating CO_2 capture ability which significantly affects the reaction efficiency. Inert CO_2 molecular in the gas phase needs to be adsorbed and activated by the surface oxygen vacancy sites and then can react with methanol to generate $DMC^{[9-12]}$. Doping metal ions while maintaining the fluorite crystalline structure of ceria is one of the effective ways to enhance the concentration of surface oxygen vacancy of $CeO_2^{[9,13-14]}$. Because the impurity ions can reduce the crystalline size, generate more surface defects and boost the reducibility of surface oxygen^[15-16]. On another hand, the surface acid - base property of CeO_2 can be mediated by the doping method, which will further favor the formation of DMC to improve the selectivity^[9]. According to Scheme 1, the reaction equilibrium can shift toward the right side by water removal^[17]. Usually, inorganic dehydrating agents are introduced to physically remove water with limited effect due to the low dehydration capacity at reaction temperatures^[18-21]. While organic dehydrating agents are applied to remove water by chemical reactions which may form lots of by-products complicating the entire process^[22-25]. Coating the catalyst powder on the surface of cordierite honeycomb ceramics can improve the phase-phase mass transfer performance^[26-28]. Therefore, it is reasonable to expect an enhanced efficiency for water removal using a honeycomb structure catalyst, which will improve the DMC yield in return.

In this contribution, $\text{Ce}_{1-x}\text{Mg}_x\text{O}_2$ (x=0.05, 0.10, 0.15, 0.20) solid solutions with a variation of magnesium content were prepared by the co-precipitation method to find an optimal ratio. Mg ions doping in CeO₂ lattice adjusted the surface acid-base property and the surface oxygen vacancies. Among all the obtained catalytic materials, Ce_{0.90}Mg_{0.10}O₂ was found to show the best catalytic activity in the direct synthesis of DMC from methanol and carbon dioxide. Using a unique structure, monolithic catalyst produced by coating powder on cordierite honeycomb ceramics showed high effective and stable catalytic performance. At 140 °C, 2.4 MPa, and 2 h continuous reaction, the yield of DMC over Ce_{0.90}Mg_{0.10}O₂ monolithic catalyst was the highest (20.21%).

1 Experimental

1.1 Materials preparations

The preparation of $Ce_{0.90}Mg_{0.10}O_2$ by the coprecipitation method is described as an example. We weighed 15.000 0 g (NH₄)₂Ce(NO₃)₆, 0.779 5 g Mg(NO₃)₂ •6H₂O, and 70.000 0 g urea (CH₄N₂O) and dissolved them completely with 500 mL deionized water under stirring. The mixture was transferred to a 1 000 mL three-neck flask and gradually heated to 90 °C under mechanical stirring (600 r·min⁻¹) for 5 h. After the reaction, the product was cooled to room temperature naturally, the precipitate was filtered and washed with water (over 4 000 mL) and absolute ethanol (about 300 mL), dried overnight at 80 $^{\circ}$ C, and calcined at 400 $^{\circ}$ C for 4 h in the air to obtain the target product. The obtained Ce_{1-x}Mg_xO₂ powder was ground with the required deionized water to obtain a slurry, which was coated on a cordierite honeycomb ceramics (64 cells per cm², Φ : 10 mm, L: 25 mm). The load was maintained at 0.5 g, and the excess slurry was blown away. Finally, the coated catalyst was dried overnight at 80 °C and calcined at 400 °C for 4 h in the air to obtain a Ce_{0.90}Mg_{0.10}O₂ monolithic catalyst. The preparation method of Ce_{0.95}Mg_{0.05}O₂, Ce_{0.85}Mg_{0.15}O₂, and Ce_{0.80}Mg_{0.20}O₂ monolithic catalysts were the same as above, only the mass of $Mg(NO_3)_2 \cdot 6H_2O$ was changed.

1.2 Catalytic tests

The catalytic activity of the prepared catalyst for the direct synthesis of DMC from CO₂ and methanol was evaluated in a continuous fixed-bed reactor. Water was the main disadvantageous factor for the formation of DMC in the synthesis reaction. The flow of the reaction system can remove the water vapor well and detect the reaction products online. A typical procedure was to place the prepared $Ce_{1-x}Mg_xO_2$ monolith catalyst in a stainless steel reaction tube. The reactor was sealed and purged with a CO_2 stream for 30 min to drain the internal air. When the reaction system reached the required temperature, a mixed gas stream of CH₃OH $(0.145 \text{ mL} \cdot \text{min}^{-1})$ and CO₂ (40 mL $\cdot \text{min}^{-1}) (n_{\text{CH}_2 \text{OH}}: n_{\text{CO}_2} =$ 2:1) was introduced. Then the reaction was carried out at 140 °C, 2.4 MPa, and 2 880 h⁻¹ of gas hourly space velocity (GHSV). The outlet component after the reaction was analyzed online using gas chromatography (Agilent 7890B) equipped with a hydrogen flame ionization detector. The calculation formula for CH₃OH conversion and DMC selectivity is as follows:

$$Conversion = \frac{2c_{DMC} + c_{HCHO} + 2c_{DME}}{c_{CH_3OH} + 2c_{DMC} + 2c_{DME} + c_{HCHO}} \times 100\%$$
(1)

Selectivity =
$$\frac{c_{\rm DMC}}{c_{\rm DMC} + c_{\rm DME} + c_{\rm HCHO} + c_{\rm CO}} \times 100\%$$
 (2)

Where c_i represents the concentration of a component (i).

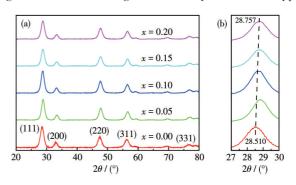
2 Results and discussion

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2.1 Characterization of as-prepared solid solutions

Fig. 1 shows the X-ray diffraction (XRD) patterns of the prepared $Ce_{1-x}Mg_xO_2$ composite oxides (Detailed characterization conditions can be found in Supporting Information). CeO₂ samples showed typical diffraction lines of cubic fluorite structure (PDF No. 43-1002). Besides, it can be seen that the catalyst doped with Mg²⁺ still maintained the characteristic peak of cubic fluorite ceria after calcination, no diffraction line representing MgO or any other impurities was detected. Compared with pure CeO_2 , the (111) plane peak shifted to a higher angle with increased Mg concentration (Fig. 1b), indicating a lattice contraction. The calculated lattice constant decreased from 0.541 8 nm for CeO₂ to 0.540 6 nm for $\mathrm{Ce}_{0.80}\mathrm{Mg}_{0.20}\mathrm{O}_2$ (Table 1) because the ionic radius of Mg^{2+} (0.089 nm) is smaller than that of Ce^{4+} (0.097 nm). The XRD patterns imply that the Mg²⁺ incorporate into the CeO₂ lattice forming no MgO species and part of them substitutes the Ce4+ leading to lattice contraction. These results are in good agreement with previous reports^[15,29-30]. The calculated grain size from (111) for all samples ranges from 5.8 to 6.1 nm and the specific surface area is basically the same, indicating that the addition of Mg has little influence on the micro-textural property.

The N_2 adsorption-desorption isotherms and pore size distributions of $Ce_{1-x}Mg_xO_2$ catalyst are shown in Fig.S1. As shown in Fig.S1, all catalysts obtained type



 $\label{eq:rescaled} \begin{array}{ll} \mbox{Fig.1} & \mbox{(a) XRD patterns of $Ce_{1-x}Mg_xO_2$ composite oxides} \\ & \mbox{and (b) zoomed-in view of the (111) plane peak} \end{array}$

W isotherms with clear H3 hysteresis lines, indicating typical mesoporous materials. In Fig. S2, all catalysts contain mesopore pore size distributions with pore sizes ranging from 2 to 20 nm. The above results show that the Mg²⁺ content has a significant effect on the pore size distribution. The BET (Brunauer - Emmett - Teller) surface area and pore volume of the synthesized $Ce_{1-x}Mg_xO_2$ catalyst are summarized in Table 1. It can be observed that $Ce_{0.90}Mg_{0.10}O_2$ composite oxide possesses the highest specific surface area of 136 m²·g⁻¹ and pore volume of 0.188 cm³·g⁻¹.

Transmission electron microscope (TEM) images (Fig.2) of as-prepared $Ce_{1-x}Mg_xO_2$ composite oxides indicated that all samples were in irregular spherical

shape exposing no specific facets. The average particle size of as - prepared $Ce_{1-x}Mg_xO_2$ is consistent with the grain size.

There are two bands observed in Raman spectra (Fig. 3). The vibration peak around 461 cm⁻¹ can be attributed to the F_{2g} vibrational mode of Ce—O, which usually shows a sharp and symmetric band at 466 cm^{-1[9,31]}. Considering the high specific surface area of the prepared material, the peak shifted to low frequency and showed asymmetric character, which are mainly attributed to the small particle size. Compared with asprepared CeO₂ nanoparticles, the F_{2g} band gradually blue-shifted with increased Mg²⁺ content, which demonstrates the decreased average length of Ce—O bond

Table 1 Structural and textural properties of Ce_{1-x}Mg_xO₂ composite oxides

Catalyst	(111) plane		T	Particle size ^b / nm	$S = (m^2 - m^{-1})$	$V = (m^3, n^{-1})$
	2 <i>θ</i> / (°)	d / nm	- Lattice parameter ^a / nm	Particle size" / nm	$S_{\rm BET} / ({\rm m}^2 \!\cdot\! {\rm g}^{-1})$	$V_{\rm Pore} / ({\rm m}^3 \cdot {\rm g}^{-1})$
CeO_2	28.510	0.312 8	0.541 8	8.6	133	0.131
$Ce_{0.95}Mg_{0.05}O_2$	28.709	0.311 3	0.540 5	6.7	129	0.142
$Ce_{0.90}Mg_{0.10}O_2$	28.730	0.310 0	0.540 1	5.8	136	0.188
$Ce_{0.85}Mg_{0.15}O_2$	28.770	0.310 1	0.539 8	6.1	117	0.129
$Ce_{0.80}Mg_{0.20}O_2$	28.757	0.310 2	0.540 6	6.3	126	0.168

^a Calculated using Vegard's law; ^b Estimated by TEM.

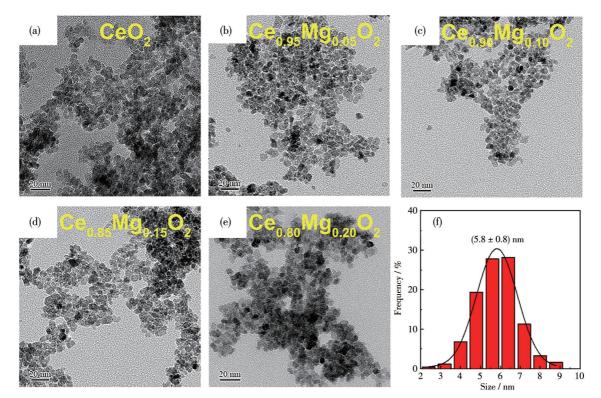


Fig.2 (a-e) TEM images of Ce1-xMgxO2 composite oxides; (f) Size distribution of Ce0.90Mg0.10O2

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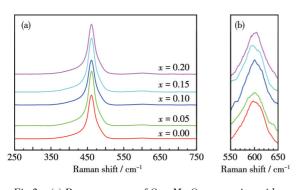


Fig.3 (a) Raman spectra of $Ce_{1-x}Mg_xO_2$ composite oxides and (b) zoomed-in view of the peak attributed to oxygen vacancies

and lattice contraction further. Therefore, it is reasonable to deduce that smaller Mg^{2+} cations substitute some Ce^{4+} ions in the fluorite lattice. It is also noted that the intensity of F_{2g} decreased with increased Mg^{2+} content, revealing structural distortion^[32-33]. Another band near 596 cm⁻¹ is related to the oxygen vacancies caused by the Ce^{3+} ion in the CeO_2 lattice (Fig. 3b)^[34]. The intensity of this mode increased with an increase of Mg^{2+} content, pointing at increased intrinsic oxygen vacancy concentration. No Raman shifts of MgO were observed in $Ce_{1-x}Mg_xO_2$, which further infers $Ce_{1-x}Mg_xO_2$ prefers a solid solution state.

To elaborate on changes in the CeO₂ chemical state after Mg doping, X-ray photoelectron spectroscopy (XPS) analysis was carried out. The XPS spectra of Ce3*d* (Fig. 4a) exhibit complex features with eight peaks. U and V represent spin - orbits of Ce3*d*_{3/2} and Ce3*d*_{5/2}, respectively. Spin - orbit doublet (V''' ca. 898.3 eV and U''' ca. 916.8 eV, V'' ca. 888.9 eV and U'' ca. 907.4 eV, V ca. 882.4 eV and U ca. 900.9 eV) are attributed to the Ce⁴⁺ species, while (V' ca. 884.9 eV and U' ca. 903.4 eV) are assigned to the Ce³⁺ species^[29,31]. Then the concentration of Ce³⁺ can be estimated by taking the ratio of the area of the integrated peak corresponding to Ce³⁺ to the total area of fitted peaks. It is shown that Mg doping has enhanced the concentration of Ce³⁺ on the surface remarkably, and the maximum ratio (19.42%) has been obtained when 10% Mg doping. The O1s XPS spectra (Fig. 4b) of Ce_{1-r}Mg_rO₂ composite oxides can be deconvoluted into 3 surface oxygen species: lattice oxygen (O_L ca. 529.3 eV), surface oxygen vacancies ($O_v ca. 530.5 \text{ eV}$); and chemisorption oxygen species (O_c) at the highest binding energy (ca. 532.2 eV)^[35]. The intensity ratio of surface oxygen vacancies to the sum of all oxygen species was summarized in Table 2. It was observed that the incorporation of Mg²⁺ can effectively increase the number of surface oxygen species (O_v+O_c) . These results confirm that there are enhanced mobility and availability of lattice oxygen species due to the synergistic effect between MgO and CeO₂.

Fig. 5 shows the temperature-programmed reduction by hydrogen (H₂-TPR) profile of as-prepared $Ce_{1-x}Mg_xO_2$ composite oxides. The TPR of pure CeO_2 showed a broad peak starting at 500 °C and one peak at 825 °C, representing the surface and the bulk reduction process, respectively. The surface reduction initiated around a lower temperature 500 °C after Mg²⁺ ions (less than 20%) were introduced, which means the reducibility of surface oxygen species has been significantly improved. Meanwhile, the area of this broad peak increased gradually with higher Mg concentration as well, indicating the lattice oxygen in bulk can move to

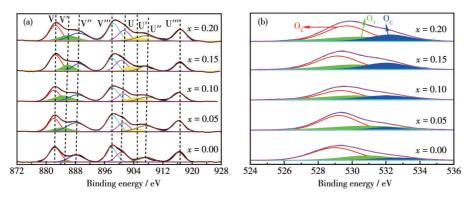


Fig.4 (a) Ce3d and (b) O1s XPS spectra of $Ce_{1-x}Mg_xO_2$ composite oxides

Catalyst	Surface content of Ce ³⁺ / %	Surface content of $O_v + O_c / \%$	Total amount of adsorbed $\text{CO}_2^{a} / (\text{mmol}_{\text{CO}_2} \cdot \mathbf{g}_{\text{cat}}^{-1})$
CeO ₂	1.34	24.97	0.787
$Ce_{0.95}Mg_{0.05}O_2$	11.57	30.85	0.794
$Ce_{0.90}Mg_{0.10}O_2$	19.42	31.98	0.845
$Ce_{0.85}Mg_{0.15}O_2$	15.18	25.06	0.808
Ce _{0.80} Mg _{0.20} O ₂	15.40	23.19	0.771

 Table 2
 Relative ratio of Ce³⁺ species and oxygen vacancies on the surface

 $^{\rm a}$ Determined using the $\rm CO_2$ temperature-programmed desorption (CO_2-TPD) analysis in Fig.S3.

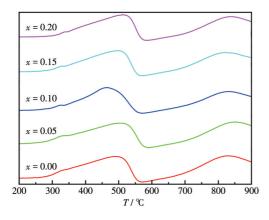


Fig.5 H_2 -TPR profiles of CeO₂ and Ce_{1-x}Mg_xO₂ composite oxides

the surface and participate in chemical reactions at a relatively lower temperature. Thus not only the reducibility of surface oxygen but also the mobility of lattice have activated due to Mg²⁺ introduction, resulting in more oxygen vacancies, probably by reducing the interaction between Ce—O with a distorted crystalline structure. This feature will facilitate chemical reactions whose reactants would be activated by oxygen vacancies. According to the related literature, the oxygen vacancy is crucial for activating carbon dioxide in the direct synthesis of DMC from CO_2 and methanol^[11,34,36].

2.2 Catalytic performance

Fig. 6a illustrates photographs of as-prepared monolithic catalyst. A scanning electron microscope (SEM) image (Fig. 6) revealed that the average thickness of the catalyst coating was ca. 60 µm. Well uniform coating layers were found, as evidenced in the corner, inner, and frontal channel views from the energy dispersion X-ray spectrum (EDS) mappings of Ce_{0.90}Mg_{0.10}O₂-coated monolithic catalyst. The abnormal distribution of Mg is due to a small amount of Mg in cordierite. It also demonstrates that Ce000 Mg01002 coated monolithic catalyst can be insufficient contact with the reaction gas stream to promote the conversion and the yield of the product^[37]. Catalyst activity of monolithic and particulate (40-60 mesh) Ce_{0.90}Mg_{0.10}O₂ catalyst was comparatively studied (Fig.7). It is easy to conclude this monolithic do have enhanced the DMC yield and methanol conversion even though both were carried out in the same fixed bed reactor. Therefore, it

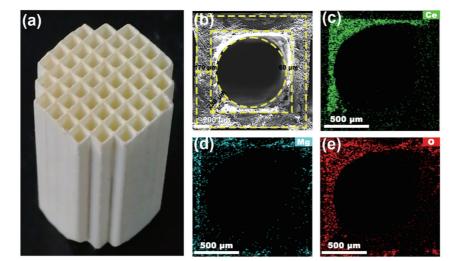


Fig.6 (a) Photographs, (b) SEM image, and (c-e) EDS element mappings on Ce_{0.90}Mg_{0.10}O₂-coated monolithic catalyst

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is probable that the unique structure of the monolithic catalyst accelerates the water removal and shifts the reaction equilibrium successfully. Fig.8 shows the performance of $Ce_{1-x}Mg_xO_2$ monolithic catalysts on direct DMC synthesis. The optimum temperature and optimum pressure can be obtained from Fig.S4 and S5. The activity of the catalyst was $Ce_{0.90}Mg_{0.10}O_2 > Ce_{0.95}Mg_{0.05}O_2 >$ $CeO_2 > Ce_{0.85}Mg_{0.15}O_2 > Ce_{0.80}Mg_{0.20}O_2$. When x=0.10, the yield of DMC reached the maximum of 20.21% and decreased with a higher doping concentration. It is mainly reflected in the decrease of DMC selectivity and the increase of HCHO and DME selectivity.

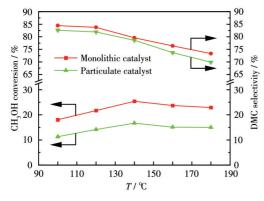
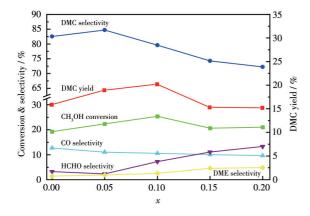


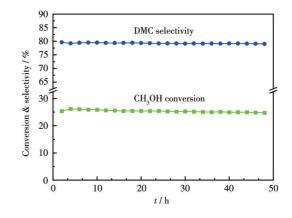
Fig.7 Catalytic activity of monolithic and particulate $Ce_{0.90}Mg_{0.10}O_2$ catalyst



Reaction conditions: catalyst: 500 mg, GHSV: 2 880 mL·g_{eat}⁻¹·h⁻¹, $n_{\rm CH_3OH}$: $n_{\rm CO_2}$ =2:1, temperature: 140 °C, pressure: 2.4 MPa

Fig.8 Catalytic performance of Ce1-xMgxO2 monolithic catalysts

According to our previous studies, there are the following reaction processes in this process: (I) $2CH_3OH \rightarrow CH_3OCH_3+H_2O$; (II) $2CH_3O+CO_2 \rightarrow HCHO+CO+H_2O^{[35]}$. It can be seen that the doping of Mg can promote the process of (I) and (II), which leads to a decrease in the selectivity of DMC.



Reaction conditions: catalyst: 500 mg, GHSV: 2 880 mL·g_{cat}⁻¹· h^{-1} , n_{CH_3OH} : n_{CO_2} =2:1, temperature: 140 °C, pressure: 2.4 MPa

Fig.9 Durability test of $Ce_{0.90}Mg_{0.10}O_2$ monolithic catalyst

To provide referable information for the industry, we examined the stability of $Ce_{0.90}Mg_{0.10}O_2$ monolithic catalyst at 140 °C and 2.4 MPa. There is little deactivation for this catalyst (DMC yield from 20.21% to 19.56%) during the 50 h durability test implies it is a quite promising application for the direct synthesis of DMC from CO_2 and methanol.

3 Conclusions

In conclusion, doping Mg in CeO₂ lattice can enhance the catalytic performance on the direct formation of DMC from methanol and CO₂. Since Mg²⁺ ions play an important role in the activation of oxygen species in CeO₂ lattice, which favors the oxygen vacancies formation. At the same time, the honeycomb structure of the monolithic catalyst greatly improves the removal of reaction products, overcoming thermodynamic limitations to some extent. Consequently, the yield of DMC and the stability of the catalyst can be improved.



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