

多次透射反射红外光谱法灵敏和准确地测量单晶硅中间隙氧和代位碳的含量

路小彬 肖守军*

(配位化学国家重点实验室, 南京大学化学化工学院, 南京 210093)

摘要: 建立了室温下使用多次透射反射红外光谱法(MTR-IR)测量单晶硅中间隙氧和代位碳含量的新红外光谱吸收方法, 在理论和实验上证明了 MTR-IR 优于常规使用的单次垂直透射红外(IR)吸收测量方法。与 IR 法相比较, MTR-IR 法的优点为: (1) 间隙氧在 $1\,107\text{ cm}^{-1}$ 处和代位碳在 605 cm^{-1} 处的吸收峰与 MTR-IR 法中红外光透过硅片的次数 $N(6\sim 12)$ 成线性增加的正比例关系, 因此单晶硅中间隙氧和代位碳含量的检测限至少比 IR 法低一个数量级; (2) MTR-IR 法测量薄硅片如 0.2 mm 的厚度时产生的干涉条纹强度是单次垂直透射红外吸收法(IR)的 $1/23$, 是单次 Brewster 角透射红外吸收法的 $1/11$; (3) 单次垂直透射红外吸收法(IR)1 次只测量样品上的 1 个点, MTR-IR 法则在更长的样品上 1 次测量多个样品点, 每次测量更具有代表性。理论计算和实验结果都证实了 MTR-IR 吸收法测量晶体硅中间隙氧和代位碳杂质含量的高灵敏度、可靠性和重复性。

关键词: 多次透射反射; 红外; 间隙氧; 代位碳

中图分类号: O611.5

文献标识码: A

文章编号: 1001-4861(2016)02-0351-09

DOI: 10.11862/CJIC.2016.044

Sensitive and Accurate Measurement of Interstitial Oxygen and Substitutional Carbon in Single Crystalline Silicon by Multiple Transmission-Reflection Infrared Spectroscopy (MTR-IR)

LU Xiao-Bin XIAO Shou-Jun*

(State Key Laboratory of Coordination Chemistry, School of Chemistry and Chemical Engineering, Nanjing University, Nanjing 210093, China)

Abstract: A new infrared spectroscopic measurement of interstitial oxygen and substitutional carbon in silicon wafers at room temperature by Multiple Transmission-Reflection Infrared Spectroscopy (MTR-IR) has been established. The superiority of MTR-IR to conventional IR has been analyzed first in principle by theoretical calculation and then verified by practical measurements of single crystalline silicon samples. The advantages of MTR-IR over conventional IR with a single normal incidence are: (1) The absorption bands of interstitial oxygen at $1\,107\text{ cm}^{-1}$ and substitutional carbon at 605 cm^{-1} can be enhanced linearly with the simplified transmission times (N) between 6 and 12, which consequently extends the detection limit of oxygen and carbon at least one order of magnitude lower. (2) The strength of interference fringes can be decreased for a 0.2 mm thin silicon slice by 23 times as that from the single normal incidence and 11 times as that from the Brewster angle transmission respectively. (3) Not like the conventional IR method, only collecting data from one sampling point at each measurement, MTR-IR collects data from multiple sampling points in a longer sample for one measurement. Overall, both theoretical calculations and experimental results demonstrate the high sensitivity, reliability, and reproducibility of the MTR-IR spectroscopy on the measurement of impurities of interstitial oxygen and substitutional carbon of single crystalline silicon.

Keywords: multiple transmission-reflection (MTR); infrared Spectroscopy (IR); interstitial oxygen; substitutional carbon

收稿日期: 2015-11-03。收修改稿日期: 2015-12-03。

国家重点基础研究发展计划(No.2013CB922101), 国家自然科学基金(No.91027019)资助项目。

*通信联系人。E-mail: sjxiao@nju.edu.cn

0 Introduction

Silicon-based semiconductor industry, as a miracle in the human beings history, continues to develop at considerably high growth rates already for half a century. Recently, solar energy has been recognized in common as an alternative sustainable energy source due to the increased awareness of the global energy crisis^[1]. Silicon solar cells have become the most important photovoltaic products owing to the sophisticated manufacturing technology and the reliable cell quality. The quality control of silicon materials is crucial to both semiconductor and solar cell industries of silicon. During the Czochralski (CZ) procedure in growing silicon single crystals for semiconductor and photovoltaic industry, oxygen and carbon are incorporated into the molten silicon to different extents from the quartz crucibles and the graphite heaters. As oxygen atoms can find sites within the lattice structure among silicon atoms, interstitial oxygen (Oi) appears. Moreover, carbon atoms occupy positions generally taken by silicon atoms in the lattice structure, and this kind of impurity is defined as substitutional carbon (Cs)^[1-3].

Interstitial oxygen and substitutional carbon are the main forms of oxygen and carbon impurities existing in silicon. Their different levels cause different physical and electrical effects in silicon, which have been extensively investigated and correlated^[4-6]. Although Oi in suitable concentrations possesses an advantage to enhance the mechanical strength of silicon because of its nailing for stretching of dislocation in the silicon lattice, higher concentrations will result in electrically active defects that decrease the performance of the devices. Thus a gettering technique has been developed to decrease Oi and avoid defects during the silicon crystal growth procedure. High content of Cs affects the way Oi nucleates and precipitates, as well as resulting in softening and breakdown of electronic components^[7-10]. Consequently, to control the silicon quality for high performance devices, it is absolutely necessary to monitor the content of Oi and Cs in silicon wafers

more accurately and sensitively.

Various techniques have been applied to determine the content of Oi and Cs in bulk silicon. These techniques consist of gas fusion analysis, secondary ion mass spectrometry, charged particle activation analysis and neutron activation analysis etc., all of which are destructive, costly and time consuming^[1]. In addition, the above methods measure the total content of elemental oxygen and carbon, including Oi and Cs and other forms of oxygen and carbon. Thus they are not specific for measuring the concentration of Oi (atoms per cm^3 , abbreviated as c_{Oi} , cm^{-3}) and of Cs (atoms per cm^3 , abbreviated as c_{Cs} , cm^{-3}). By contrast, infrared analysis is specific to c_{Oi} and c_{Cs} in silicon. Silicon atoms form bonds with the adjacent oxygen atoms (Si-O-Si) and carbon atoms (Si-C) in the lattice structure^[11-13]. The interstitial oxygen is in the form of Si-O-Si, giving three vibration peaks centered at $1\,107\text{ cm}^{-1}$ (strong), 513 cm^{-1} (medium) and $1\,718\text{ cm}^{-1}$ (weak) respectively. Among the three bands, the strongest one at $1\,107\text{ cm}^{-1}$ is usually used to determine c_{Oi} ^[14-16]. The substitutional carbon in the form of Si-C gives a vibration band at 605 cm^{-1} . However the Si-C band overlaps with the strong silicon lattice vibration (Si-Si) at 605 cm^{-1} , which renders the measurement of c_{Cs} delicate. The amount of light absorbed by Si-O-Si and Si-C is proportional to the concentration of atoms forming the bonds. Thus their corresponding infrared bands are measured and manipulated to quantitate c_{Oi} and c_{Cs} respectively. The calculation procedure is composed of subtraction of a reference (free of Oi and Cs) absorption band from a samples band, and subsequently calculation of c_{Oi} and c_{Cs} using the following formulae generally.

$$A = \alpha b \quad \alpha = \varepsilon c$$

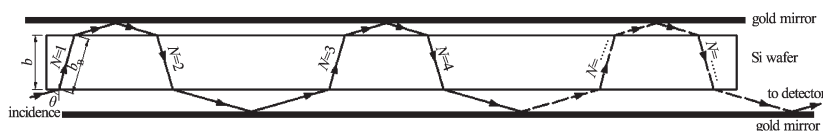
where A is the measured absorbance; ε is the absorption coefficient of a particular bond, cm^2 ; b is the thickness of the sample, cm ; c is the concentration of the impurity, cm^{-3} .

The IR method for the levels of Oi and Cs can be carried out at ambient or low temperature. Although the latter is more accurate than the former, it is costly and time consuming to handle the measurement at the

cryogenic temperature. Furthermore, the reflection loss of infrared light at both cryostat windows outweighs its advantages^[17-20].

The room temperature IR method for c_{Oi} and c_{Cs} measurements is most commonly used in industry because it is easy and simple to operate. Generally the presently accepted standard method used in industry for c_{Oi} and c_{Cs} measurements is restricted to a silicon slice with a thickness of 2 mm and a diameter of the infrared beam by a one-time perpendicular transmission of the infrared beam (we name it “conventional IR”). The detection limits of c_{Oi} and c_{Cs} for a 2.0 mm thick single crystalline silicon are 1×10^{16} and $5 \times 10^{15} \text{ cm}^{-3}$ respectively^[21-22]. The above quantitative analysis meets challenges in practice such as: 1) measurement errors become larger when c_{Oi} and c_{Cs} reach their own detection limits, 2) interference fringes from thin wafers below 0.3 mm thickness interfere the target band signal, 3) the single

transmission measurement approach in a spot provides a localized spectrum with poor specimen statistics, compared to the multi-spot data collection in MTR-IR. Our newly developed MTR-IR (Scheme 1) provides an excellent quantitative approach for analysis of c_{Oi} and c_{Cs} , due to its up to one order of magnitude enhancement of infrared absorption signals on the same silicon slice from the conventional IR measurement^[23-26]. In this letter, we applied the MTR-IR spectroscopy in analysis of c_{Oi} and c_{Cs} of 0.45 mm thick silicon wafers. Our MTR-IR method greatly improves the measurement sensitivity and accuracy, for example, reaching detection limits of c_{Oi} at $1 \times 10^{15} \text{ cm}^{-3}$ and c_{Cs} at $5 \times 10^{14} \text{ cm}^{-3}$ for the standard silicon specimens with a 2.0 mm thickness, which is one order of a magnitude lower than from the standard method. MTR-IR also attenuates the interference fringes of thin wafers greatly, and advances the representativeness of data collections.



The incident light, depicted with arrows, transmits the silicon chip N times and reflects on two gold mirrors N times too, finally reaches the detector. The parameters used in the calculation are marked: b , the thickness of the chip; θ , the Brewster incidence angle of 74° ; b_b , the optical path at the Brewster incidence^[27-28].

Scheme 1 Scheme of the MTR-IR optical path

1 Experimental

1.1 Substrates

Double-side-polished and $\langle 100 \rangle$ oriented n-type silicon wafers (B doped, resistivity of $15 \Omega \cdot \text{cm}$ (CZ) and $3\,000 \Omega \cdot \text{cm}$ (FZ) respectively, 0.45 mm thick, from Shanghai Junhe Electronic Materials Co. Ltd., China) were cut into rectangular shapes ($16 \text{ mm} \times 50 \text{ mm}$) for infrared analysis. CZ silicon wafers were used as samples and FZ as reference to measure the impurity concentrations of Oi and Cs respectively.

1.2 Wafer cleaning

Silicon wafers were cleaned with “piranha solution” (concentrated $\text{H}_2\text{SO}_4/30\% \text{H}_2\text{O}_2$, 3:1, V/V) for 4 h (caution: piranha solution reacts violently with

organic materials and should be handled with great care) to remove organic pollutants, followed by boiling in the mixture of $\text{NH}_3 \cdot \text{H}_2\text{O}/\text{H}_2\text{O}/\text{H}_2\text{O}_2$ (1:1:1, V:V:V) for 30 min, then cooling to room temperature, rinsing with water, and storing in water. Silicon samples were immersed in 1% HF for 5 min to eliminate the native passivation silicon oxide layer, and dried with a stream of nitrogen just before the measurement.

1.3 Measurement

The optical setup was designed to adapt to any commercial FTIR spectrometers, which is Bruker 80v FTIR, in our case. MTR accessory with a Brewster incident angle of 74° was used. Unless specified, a DTGS detector and scan times of 100 at 4 cm^{-1} resolution were used for measurement over the

wavenumber range from 400 to 4 000 cm^{-1} .

The silicon sample was inserted between the two Au mirrors, with one end of the silicon slice protruding about 5 mm out of the incident spot, in order to make sure that the first incidence shot was on the silicon surface. The two guiding mirrors can be moved back and forth to get the maximal luminous flux in the DTGS detector. The incident angle was controlled at 74° by a micro-adjuster with a minimal angle scale of 0.225.

The whole measurement procedure was performed according to ASTM F 1188 and ASTM F 1391^[21-22,30]. Eight random samples from different batches were used. Each sample was measured successively 4 times by MTR and IR by slightly relocating the Si wafer each time in order to measure different sampling points. Therefore for each sample, 4 different sampling points were measured by IR, whereas 40 different sampling points measured by MTR-IR if N equals 10.

1.4 Theory/calculation

1.4.1 Comparison of computation models for conventional IR and MTR-IR

1.4.1.1 Conventional IR method

In the conventional IR method, the normal incident light passes a Si wafer (for convenience, a slightly oblique incidence is drawn in Fig.1), the transmittance can be expressed as in Eq.(1) and (2)^[21-22].

$$T = \frac{(1-R)^2 \exp(-\alpha b)}{1 + R^2 \exp(-2\alpha b) - 2R \cos(2\psi) \exp(-\alpha b)} \quad (1)$$

$$\psi = 2\pi n b \sigma \quad (2)$$

where T is the transmittance of normal incidence, %; R is reflectivity; n is refractive index; σ is wavenumber, cm^{-1} ; ψ is the phase change due to the interfering multiple reflections on the boundaries of the sample.

1.4.1.2 Brewster angle single incidence

The expression of T_B for the single Brewster incidence refers to Eq.(3)^[29].

$$T_B = \frac{1}{2} \exp(-\alpha b_B) + \frac{1}{2} \frac{(1-R)^2 \exp(-\alpha b)}{1 + R_s^2 \exp(-2\alpha b_B) - 2R_s \cos(2\psi) \exp(-\alpha b_B)} \quad (3)$$

where T_B is the transmittance of the Brewster angle single transmission, %; R_s and R_p indicate the reflectance of s - and p -polarization respectively; the first ($R_p=0$) and the second term (R_s has a value) in the formula represent transmittance energy of p -polarization and s -polarization respectively, because the total transmittance energy equals the sum of p -polarization and s -polarization (p - or s -polarization holds 1/2 of the original light energy); the phase change introduced by the interfering multiple reflections on the boundaries of a silicon is considered.

1.4.1.3 MTR method^[27]

The light path of the computation model for MTR-IR refers to Scheme 1. Comparing Scheme 1 to Fig.2, we observed that two gold mirrors enforce the light transmit through the silicon slice N times and simultaneously taking away the message of oxygen and carbon of the silicon sample. The optical path in the MTR-IR setup of Scheme 1 is N times of b_B at the

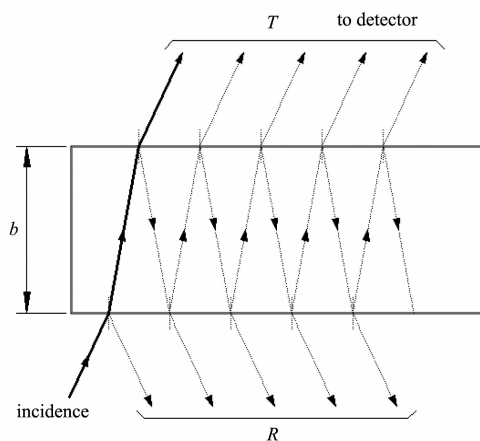


Fig.1 Light path of computation model for IR

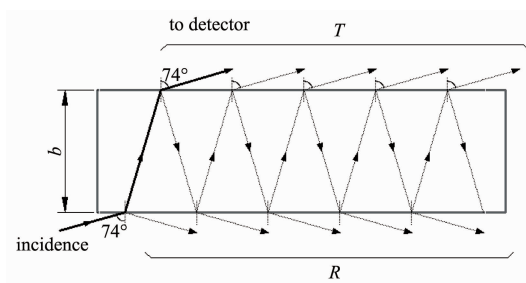


Fig.2 Light path of the computation model for Brewster angle single transmission

Brewster incidence. For simplicity, only the main light path is drawn in Scheme 1 with $N=6$, all other light paths by multiple reflections in the MTR setup are ignored. The transmittance for MTR-IR (T_{MTR}) is deduced as follows (See supporting information for details of derivation of the formula of MTR-IR):

$$T_{\text{MTR-O}} = \frac{\exp(-\alpha_{\text{O}} b_{\text{MTR}})}{2} R_{\text{Au}}^{N-1} [1 + \frac{(1-R_s)^2}{1+R_s^2 \exp(-2\alpha_{\text{O}} b_{\text{B}}) - 2R_s \cos(2\psi_{\text{O}}) \exp(-\alpha_{\text{O}} b_{\text{B}})}]^N \quad (4)$$

$$T_{\text{MTR-O}} = \frac{\exp(-\alpha_{\text{C}} b_{\text{MTR}})}{2} R_{\text{Au}}^{N-1} [1 + \frac{(1-R_s)^2}{1+R_s^2 \exp(-2\alpha_{\text{C}} b_{\text{B}}) - 2R_s \cos(2\psi_{\text{C}}) \exp(-\alpha_{\text{C}} b_{\text{B}})}]^N \quad (5)$$

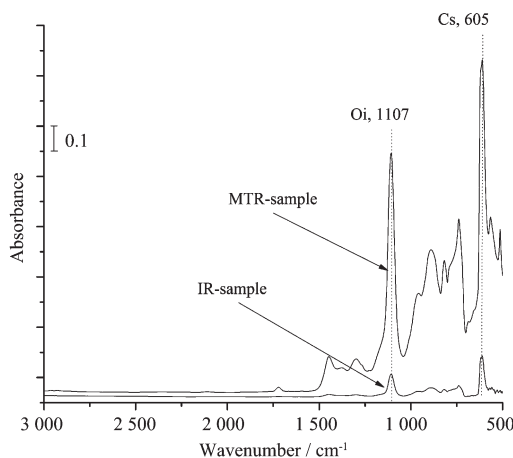
where T_{MTR} is the transmittance of MTR, %; R_{Au} is the reflectivity of gold; b_{MTR} is the optical path through a silicon sample in the MTR-IR measurement, cm.

1.4.2 Theory of signal enhancement of MTR-IR

Presently, the conventional IR method for measurement of c_{O_i} and c_{C_s} is to place a 2 mm thick

single crystalline silicon slice normally (or within the margin of error of a slightly oblique angle $\leq 10^\circ$) in the light path. The sampling length equals to the thickness of the slice. The illuminated area is a circle with a diameter of the light spot, so the resulting c_{O_i} and c_{C_s} just represent the impurities within a cylinder with a diameter of the light spot (depending on the aperture) and a height of 2 mm. Whereas in the MTR-IR setup, the infrared light reflects back and forth many times between two gold mirrors and simultaneously passes through the sampling silicon slice repeatedly and takes away the information of O_i and C_s by the resonance of Si-O-Si and Si-C bonds. So, the optical path is $N(1+1/n^2)^{1/2}$ times of the thickness of a silicon wafer (b), and correspondingly the absorbance is amplified $N(1+1/n^2)^{1/2}$ times^[29].

At the Brewster angle of 74° , when $R_p=0$, $R_s=0.70$, $R_{\text{Au}}=1$, the phase change of p polarization light $\psi=\pi$, the following formula can be deduced from Eq.(1) and



Length of the sample is 50 mm, $n=3.42$, $N=10$, $b=0.045$ cm, $\theta=74^\circ$. “MTR-sample” denotes the spectrum of a sample measured with the MTR-IR method; “IR-sample” indicates the spectrum of the same sample measured with the conventional IR method

Fig.3 Comparison of O_i and C_s spectra for MTR-IR and conventional IR

(4).

$$b_{\text{MTR}} = Nb(1 + \frac{1}{n^2})^{1/2} \quad (6)$$

$$A_{\text{IR}} = \alpha b$$

$$A_{\text{MTR-IR}} = \alpha b_{\text{MTR}} \Rightarrow \frac{A_{\text{MTR-IR}}}{A_{\text{IR}}} = \frac{\alpha b_{\text{MTR}}}{\alpha b} = \frac{\alpha Nb(1 + \frac{1}{n^2})^{1/2}}{\alpha b} = N(1 + \frac{1}{n^2})^{1/2} \quad (7)$$

From Eq. (6), the sampling length is enlarged $N(1 + 1/n^2)^{1/2}$ times, assuming the recorded infrared spectral signal comes from the main optical path illustrated in Scheme 1, and all other infrared signals are ignored. The infrared light passes through a silicon slice N times in different regions, thus the collected signal is an integrated one of the whole optical path, physically and statistically representing Oi and Cs in the silicon slice better.

2 Results and discussion

2.1 Comparison of infrared traces between the conventional IR method and the MTR-IR method

We listed two spectral traces of the same sample in Fig.3, where the upper trace was obtained from MTR-IR and the lower trace from the conventional IR. All bands bear the same shape but their absorbance strength in MTR-IR is much higher than from IR, especially for the two strongest bands of Oi at $1\ 107\ \text{cm}^{-1}$ and Cs at $605\ \text{cm}^{-1}$. From the view point of quantitation, the bigger the absorbance value, the less the measurement error of Oi and Cs. Consequently c_{Oi} and c_{Cs} are more accurate and will have a lower detection limit. Because the absorption of Cs at $605\ \text{cm}^{-1}$ overlaps with the strongest absorption of the silicon lattice vibration (Si-Si) centered at $610\ \text{cm}^{-1}$, a FZ silicon reference is needed to subtract the silicon lattice vibration. Further, the real advantage of MTR-IR over the conventional IR not only lies in its ability to measure the spectra of Oi and Cs with stronger signals, but also with more sampling points for robust and representative measurements.

In Fig.3, the peak height ratios of Oi and Cs (MTR-sample/IR-sample) were measured to be ~ 10

and ~ 8 respectively. Generally speaking, c_{Cs} is more difficult to be measured than c_{Oi} using the conventional IR method, due to two factors: (1) It is very tough to extract the much smaller Si-C peak from the strong Si-Si lattice band and therefore artificial results are often derived individually, thus an accurate quantitation is nearly impossible. (2) The Cs level in single crystalline silicon is always an order of magnitude lower than that of Oi. Judged from the signal enhancement, it is possible to extend the limit of detection of c_{Oi} at 1×10^{16} to $1 \times 10^{15}\ \text{cm}^{-3}$ and c_{Cs} at 5×10^{15} to $5 \times 10^{14}\ \text{cm}^{-3}$ for a 2.0 mm thick single crystalline silicon.

We randomly chose 8 samples to measure their c_{Oi} and c_{Cs} , both by the conventional IR and the MTR-IR methods, for verification of the MTR-IR method by the correlation curve in Fig.4a and 4b (calculation details please see Supporting Information: 2. Data calculation). As it can be seen in Fig.4a and 4b, both c_{Oi} and c_{Cs} are linearly correlated. The linear relationship of both c_{Oi} and c_{Cs} proved the measurement accuracy of the MTR-IR method for determination of c_{Oi} and c_{Cs} in single crystalline silicon materials.

The interference fringes become much stronger when a silicon wafer thickness is close to the infrared wavelength. The fringes are well recognized to interrupt the IR measurement, they obscure the weak features of the spectra, as well as reduce the accuracy

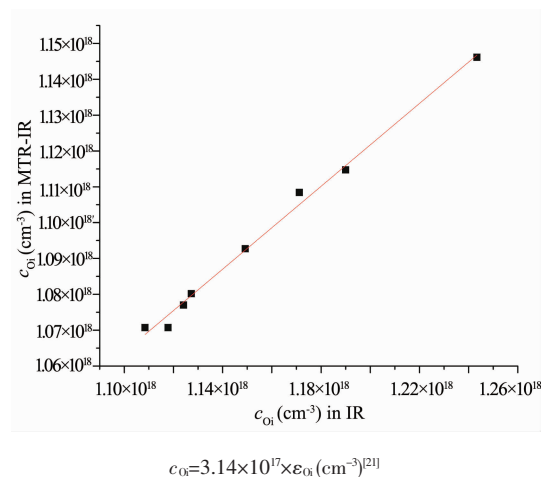


Fig.4a Oi concentration at $1\ 107\ \text{cm}^{-1}$ for eight samples measured by MTR-IR vs conventional IR

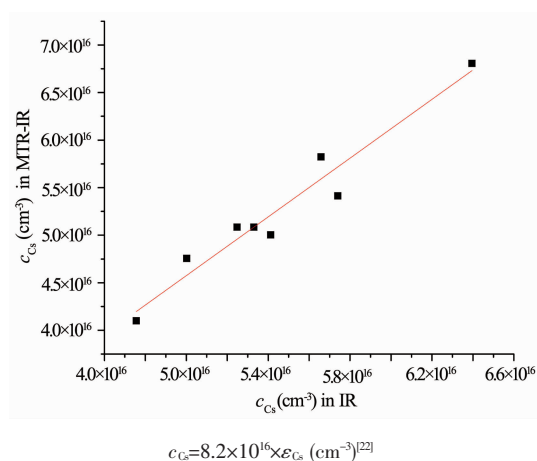
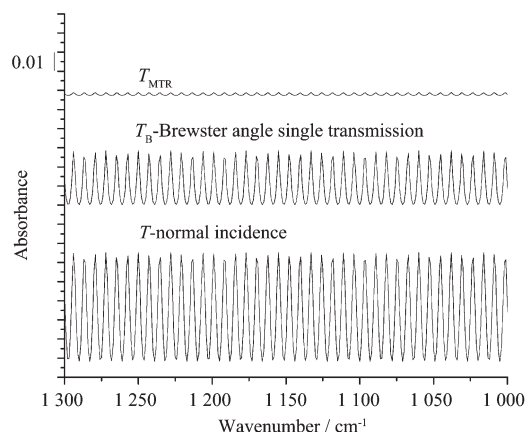


Fig.4b Cs concentration at 605 cm^{-1} for eight samples measured by MTR-IR vs conventional IR

of quantitative analysis. Both IR and MTR-IR are able to measure a silicon slice with a thickness above 0.3 mm because the multiple beams resulted from multiple reflections and transmissions are out of phase and thus these beams generate neglectable interference fringes^[29].

However, when the sample thickness is less than 0.2 mm, the interference fringes become more obvious and cannot be ignored when measuring Oi and Cs. The amplitude of interference fringes depends on the interaction mode between the incident light and the sample. Eq.(1), (3) and (4) are used to calculate the transmittance of three modes respectively: normal incidence, Brewster angle single transmission, and MTR. For the Brewster angle incidence, R_p is equal to 0, while R_s is 0.70. Obviously the oscillation of transmittance is derived from the phase change of ψ , thus the p -polarization does not cause any oscillation of transmittance at the Brewster angle incidence. In (1), the whole term affects the amplitude of oscillation, while in (3) and (4), the second term bearing the phase change of ψ becomes weaker and weaker when N increases. Therefore the amplitude of oscillations must be reduced with increasing N . In Fig.5, the variation of oscillations caused by phase change (ψ) is shown for the normal incidence, the Brewster angle single transmission, and the MTR mode respectively. The amplitude of the oscillation is 0.058 for the normal incidence (T), 0.029 for the Brewster angle



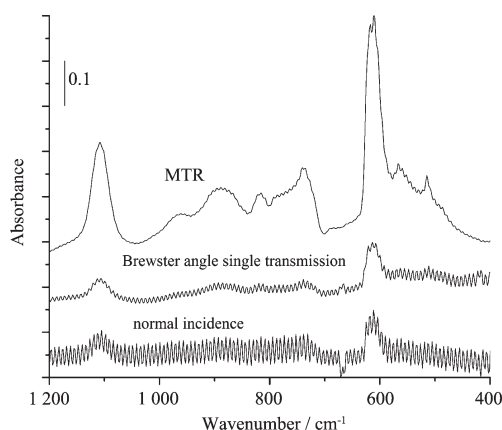
With assignment of $n=3.42$, $\varepsilon=39.5 \text{ cm}^2$, $b=0.02 \text{ cm}$, $b_B=0.0208 \text{ cm}$, $b_{MTR}=0.1664 \text{ cm}$, $\theta=74^\circ$, $N=8$

Fig.5 Theoretical simulation of the transmittance oscillations as a function of the wavenumber σ calculated from Eq.(1), (3) and (4), corresponding to the normal incidence, the Brewster angle single transmission and MTR respectively

single transmission (T_B), and only 0.0025 for the MTR-IR approach (T_{MTR}). In this case, theoretically compared to the normal incidence, the Brewster angle single transmission reduces the oscillation amplitude by a factor of 2, whereas, the MTR setup reduces the oscillation amplitude by a factor of 23. The above theoretical analysis demonstrates the overwhelming advantages of the MTR setup for measuring c_{Oi} and c_{Cs} in a thin silicon slice.

To prove the theoretical calculation, we recorded the spectra of a 0.20 mm thin silicon wafer in Fig.6 with normal incidence (bottom trace as “normal incidence”), Brewster angle single transmission (middle trace as “Brewster angle single transmission”) and MTR (upper trace as “MTR”) respectively. From the three curves, it is easily observed that interference fringes appear heavily for the normal incidence, moderately for the Brewster angle single transmission, and negligibly for the MTR. The interference fringe strength ratios of MTR/Brewster angle single transmission/normal incidence are close to 20:2:1 in most regions.

From Fig.6, not only the interference fringes in MTR-IR are greatly attenuated, the target signals in



With a sample length of 5 cm, $n=3.42$, $N=8$, $b=0.02$ cm, $b_B=0.0208$ cm, $b_{MM}=0.1664$ cm, $\theta=74^\circ$

Fig.6 Experimental results of interference fringes corresponding to normal incidence, Brewster angle single transmission and MTR respectively

the region of 400 to 1 200 cm^{-1} are also significantly magnified. Thus the signal to noise ratio is enhanced several orders of magnitude higher in MTR for much thinner silicon slices less than 0.3 mm thickness. The use of a thin silicon wafer can decrease the sampling volume, save the cost of an experiment, and fits the requirements of the solar energy industry. From both the theoretical calculations of Fig.5 and the experimental spectra of Fig.6, obviously the currently

used standard IR method is not suitable, whereas MTR-IR is much more powerful for measurement of Oi and Cs in a thin silicon slice less than 0.3 mm thick. That is also why MTR-IR is needed urgently for the silicon solar cell industry.

Finally, we present the evolution of 7 infrared traces against N in Fig.7a. We have demonstrated the linear relationship of the absorbance strength of a band against the number of simplified transmission times (N) in our previous report^[23]. Since the detection limit of an analysis method depends on the signal to noise ratio. For the measurements of Fig.3 and Fig.7a, their baselines are flat enough, therefore, we can assign the spectral noise from 0.3~2.0 mm thick silicon samples to the instrumental noise, which possess the same value. With this hypothesis, we can deduce the detection limits of Oi and Cs, shown in Fig.7b, at different N from 6 to 12 by dividing the standard detection limits with the peak magnification times of MTR/normal incidence (peak height of Oi or Cs in Fig.7 /peak height of Oi or Cs in IR-sample in Fig.3). It is observed that our MTR-IR method significantly improves the detection limits of c_{Oi} and c_{Cs} for the standard 2.0 mm thick single crystalline silicon, reaching c_{Oi} and c_{Cs} at 1×10^{15} and $5 \times 10^{14} \text{ cm}^{-3}$, respectively.

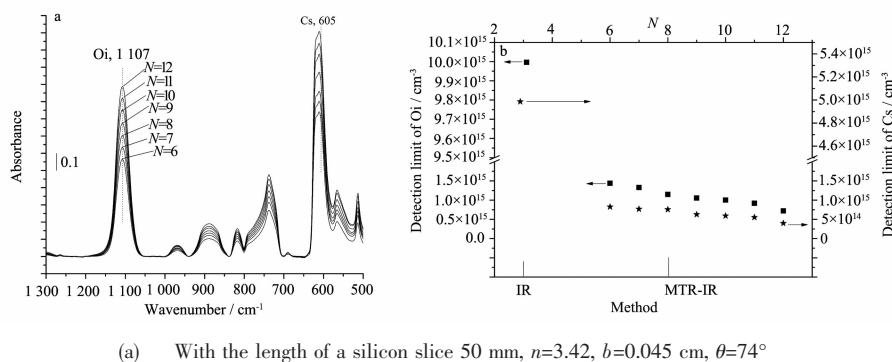


Fig.7 (a) Variation of the absorbance of Oi and Cs in silicon wafers vs N from the MTR setup (b) Comparison of the limit of detection of c_{Oi} and c_{Cs} in silicon wafers measured with IR and MTR-IR

3 Conclusions

In conclusion, our experiments confirm that the MTR-IR method can reach a higher sensitivity and better spectral quality than the most commonly used conventional IR. The signal of the Oi peak at 1 107

cm^{-1} obtained by MTR can be enhanced 10 times than by the conventional IR method, the one of the Cs peak at 605 cm^{-1} 8 times. The MTR sampling length is $N(1+1/n^2)^{1/2}$ times long as the one in the conventional IR, thus the measured c_{Oi} and c_{Cs} are more representative. Since MTR-IR reduces the

interference fringes greatly for silicon slices with a thickness thinner than 0.3 mm, it will be the most powerful tool to characterize the ultrathin silicon wafers and therefore the portable and foldable silicon devices. Due to its simple operation, MTR-IR satisfies the practical needs in industrial applications, especially for semiconductor and silicon solar cell industries.

Several parameters still need improvement in further works. For example, the theoretical equations are deduced from the main light path, neglecting other multiple reflections and transmissions on silicon and gold mirrors. The MTR formulas still need more experimental data for calibration.

Considering the conclusion above, we believe that the MTR-IR method will be established as a standard method for measurement of interstitial oxygen and substitutional carbon for crystalline silicon materials.

Acknowledgments: We acknowledge financial support from the National Basic Research Program of China (No. 2013CB922101) and the NSFC, No. 91027019.

Supporting information is available at <http://www.wjhxxb.cn>

References:

- [1] Boyle R. *Thermo Scientific Application Note*, **2008**,**50640**:1-4
- [2] Craven R A, Korb H W. *Solid State Technol.*, **1981**,**24**(7):55-61
- [3] Benson K E, Lin W, Martin E P. *Semiconductor Silicon* 1981. Pennington N.J.: Electrochem. Soc. Inc., **1981**:33-48
- [4] Abe T, Kikuchi K, Shirai S, et al. *Semiconductor Silicon* 1981. Pennington N.J.: Electrochem. Soc. Inc., **1981**:54-71
- [5] Rava P, Gatos H C, Lagowski J. *Semiconductor Silicon* 1981. Pennington N.J.: Electrochem. Soc. Inc., **1981**:232-243
- [6] Ohsawa A, Honda K, Yoshikawa M. *Fujitsu Scie. Techn. J.*, **1980**,**16**(3):123-134
- [7] Kishino S, Matsushita Y, Kanamori M. *Appl. Phys. Lett.*, **1979**,**35**(3):213-215
- [8] Ogino M. *Appl. Phys. Lett.*, **1982**,**41**(9):847-849
- [9] Oehrlein G S, Lindstrom J L, Corbett J W. *Appl. Phys. Lett.*, **1982**,**40**(3):241-243
- [10] Ohsawa A, Takizawa R, Honda K, et al. *Appl. Phys.*, **1982**, **53**(8):5733-5737
- [11] Pajot B. *Analysis.*, **1977**,**5**:293-303
- [12] Hrostowski H J A B J. *J. Chem. Phys.*, **1960**,**33**:980-990
- [13] Corbett J W, McDonald R S, Watkins G D. *J. Phys. Chem. Solids*, **1964**,**25**:873-879
- [14] Kaiser W, Keck P H, Lange C F. *Phys. Rev.*, **1956**,**101**(4): 1264-1267
- [15] Kaiser W, Keck P H. *J. Appl. Phys.*, **1957**,**28**(8):882-885
- [16] Kaiser W, Frisch H L, Reiss H. *Phys. Rev.*, **1958**,**112**(5): 1546-1554
- [17] Bosomworth D R, Hayes W, Spray A R L, et al. *Royal Soc. London*, **1970**,**317**(1528):133-152
- [18] Pajot B, Deltour J P. *Infrared Phys.*, **1967**,**7**:195-200
- [19] Oeder R, Wagner P. *Defects in Semiconductors II*. N.Y.: North-Holland, **1983**:171-175
- [20] Kolbesen B O, Kladenovi T. *Krist. Tech.*, **1980**,**15**(1):k1-k3
- [21] ASTM. *Designation F1188*: Test Method for Interstitial Atomic Oxygen Content of Silicon by Infrared Absorption.
- [22] ASTM. *Designation F1391*: Test Method for Substitutional Atomic Carbon Content by Infrared Absorption.
- [23] Liu H, Xiao S, Chen Y, et al. *J. Phys. Chem. B*, **2006**,**110** (36):17702-17705
- [24] Guo P, Liu H, Liu X, et al. *J. Phys. Chem. C*, **2010**,**114**(1): 333-341
- [25] Liu H, Venkataraman N V, Bauert T E, et al. *J. Phys. Chem. A*, **2008**,**112**(48):12372-12377
- [26] Liu H, Venkataraman N V, Spencer N D, et al. *Chemphyschem*, **2008**,**9**(14):1979-1981
- [27] LIU Hong-Bo (刘洪波). *Thesis for the Doctorate of Nanjing University*(南京大学博士论文). **2008**.
- [28] Xiao S, Liu H, Tobias B. *China Patent*, **2006**,10097859.4. 2006-11-16.
- [29] Lerouelle J. *Appl. Spectrosc.*, **1982**,**36**(2):153-155
- [30] Baghdadi A, Bullis W M, Croarkin M C, et al. *J. Electrochem. Soc.*, **1989**,**136**(7):2015-2024